# Synthesis and crystal structure refinement of cubic $\mathrm{Mg}_{6.8} \mathrm{Y}$ 

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#### Abstract

The crystal structure of the Mg -rich compound in the $\mathrm{Mg}-\mathrm{Y}$ system was determined on a single crystal. Refinement showed its composition to be $\mathbf{M g}_{6.8}-\mathrm{Y}$. The crystal structure is a variant of the cubic $\alpha$ - Mn -type (space group 143 m ). In contradistinction to a previous structure report we found that the $8 c$ site shows mixed occupancy, whereas the $2 a$ site is exclusively occupied by Y .


Keywords: Intermetallic compound; Crystal structure; Mg-Y compound

## 1. Introduction

The magnesium-yttrium phase diagram [1] contains three intermetallic compounds: $\mathrm{MgY}, \mathrm{Mg}_{2} \mathrm{Y}$, and a phase called $\mathrm{Mg}_{24} \mathrm{Y}_{5}$. The latter forms peritectically and crystallizes with a substitution variant of the cubic $\alpha$-Mn-type structure (space group $\overline{4} 3 \mathrm{~m}$, Pearson code cI58). The composition $\mathrm{Mg}_{24} \mathrm{Y}_{5}(82.8 \mathrm{at} . \% \mathrm{Mg}$ ) refers to a fully ordered structure in which magnesium occupies two $24 g$ sites, and yttrium the $8 c$ and $2 a$ sites. The true structure, however, is partly disordered and the composition $\mathrm{Mg}_{24} \mathrm{Y}_{5}$ is outside the experimentally determined homogeneity range which extends from about 84 to 87 at. $\% \mathrm{Mg}$ at $525^{\circ} \mathrm{C}$ [2].

A single-crystal X-ray structure refinement [2] at the magnesium-poor phase limit suggested that the site $2 a$ has mixed occupancy, corresponding to the composition 83.6 at. $\% \mathrm{Mg}$. However, estimated standard deviations and agreement indices were not given. Apart from that study, no other single-crystal structure refinements on this compound or other magnesiumrich intermetallics having similar structures and atomic ratios have appeared in the literature [3]. We therefore decided to carry out an X-ray investigation on an $\mathrm{Mg}_{24} \mathrm{Y}_{5}$ crystal at the magnesium-rich phase limit.

## 2. Experimental

A sample was synthesized from the elements at the nominal composition $88 \mathrm{at} . \% \mathrm{Mg}$ and $12 \mathrm{at} . \% \mathrm{Y}$ by
induction melting in an alumina crucible under an argon pressure of 4 bar in order to prevent evaporation. It was annealed at $525^{\circ} \mathrm{C}$ for 2 weeks in a sealed quartz tube. X-ray powder diffraction confirmed the presence of the cubic $\alpha$-Mn-type phase with a refined cell parameter of $a=11.2507(2) \AA$. Optical micrography showed the presence of about 5 vol. $\%$ secondary phase which was not seen on the X-ray pattern. Microprobe analysis (EDAX) indicated that the composition of the main phase was $87(2)$ at. $\% \mathrm{Mg}$ and $13(2) \mathrm{at} . \% \mathrm{Y}$, and that the secondary phase was a solid solution of approximate composition $96 \mathrm{at} . \% \mathrm{Mg}$ and $4 \mathrm{at} . \% \mathrm{Y}$. The composition and cell parameter of the main phase are consistent with those reported for the magnesium-rich boundary of the $\mathrm{Mg}_{24} \mathrm{Y}_{5}$ phase [2].

A single crystal of irregular shape (mean diameter of about $80 \mu \mathrm{~m}$ ) suitable for X-ray diffraction analysis was selected from the crushed alloy and mounted on a Philips PW1100 automatic four-circle diffractometer (Mo $\mathrm{K} \alpha$ radiation, graphite monochromator). During the first stages of structure refinement the intensities of the Friedel pairs were not merged, yielding an absolute structure parameter which suggested that the set of positional parameters chosen was the correct one. Only one atom site, e.g. $8 c$, was found to have mixed occupancy ( $\mathrm{Y} / \mathrm{Mg} \sim 2$ ). No evidence for mixed or partial occupancy was found for the other sites (error limits about $1 \%$ ), in particular for site $2 a$, which was previously identified [2] as a mixed atom site. Thus their population parameters were fixed at unity,
and the sum of the occupancies of the mixed atom site $8 c$ was constrained to $100 \%$. The refined composition of that model, $\mathrm{Mg}_{6.8} \mathrm{Y}$ ( $87.2 \mathrm{at} . \% \mathrm{Mg}, 12.8 \mathrm{at} . \% \mathrm{Y}$ ), was in good agreement with the results of the microprobe analysis. The parameters of data collection and structure refinement are given in Table 1. All calculations were performed with the xtal 3.2 system [4]. The structural results are summarized in Table 2 and the interatomic distances in Table 3.

As expected, our refinement results differ significantly from those reported previously [2]. In the presently investigated magnesium-rich crystal the $8 c$ site shows mixed occupancy and the $2 a$ site is occupied by yttrium only, whereas in the previously investigated magnesium-poor crystal the $2 a$ site shows mixed occupancy and the $8 c$ site is occupied by yttrium only. Notice that a hypothetical substitution of yttrium by magnesium on site $2 a$ only would account for a compositional range $\left(\mathrm{Mg}_{4.8} \mathrm{Y}-\mathrm{Mg}_{6.25} \mathrm{Y}\right)$ which is outside the observed magnesium-rich phase limit $\left(\mathrm{Mg}_{6.8} \mathrm{Y}\right)$. It is therefore not surprising that the $8 c$ site in our magnesium-rich crystal shows mixed occupancy. What is surprising, however, is the fact that in this crystal all excess magnesium (with respect to the ideal composition $\mathrm{Mg}_{24} \mathrm{Y}_{5}$ ) goes on that site at the expense of site $2 a$ which shows mixed occupancy in the mag-nesium-poor crystal. In view of this unusual behaviour, and considering the absence of error estimates in the previous structure work [2], a re-investigation of the

Table 1
Data collection and refinement parameters

| $(\sin \theta / \lambda)$ range $\left(\AA^{-1}\right)$ | $0.089-0.725$ |
| :--- | :--- |
| $h k l$ range | $-16 \leqslant h \leqslant 16$ |
|  | $-11 \leqslant k \leqslant 11$ |
| Collected reflections | $-11 \leqslant l \leqslant 16$ |
| Independent reflections | 1044 |
| Linear absorption coefficient $\left(\mathrm{mm}^{-1}\right)$ | 261 |
| $R_{\text {int }}$ | 7.9 |
| Number of reflections with $l_{\text {rel }}>1.5 \sigma\left(l_{\text {rel }}\right)$ | 8.98 |
| Number of parameters | 200 |
| Final residual electron density $\left(\AA^{-3}\right)$ | 18 |
|  | 1.7 |
| Goodness of fit | -3.1 |
| $R(F)(\%)$ | 1.43 |
| $R_{w}(F), w=1 / \sigma^{2}\left(F_{\text {rel }}\right)(\%)$ | 6.35 |
|  | 4.42 |

Table 3
Interatomic distances shorter than $4 \AA$

| Y | -12 Mg 2 | $3.470(3)$ | M | -3 Mg 1 | $3.203(3)$ |
| :--- | :--- | :--- | :--- | :--- | :--- |
|  | -4 M | $3.633(1)$ |  | -6 Mg 2 | $3.463(3)$ |
|  |  |  |  | -3 Mg 2 | $3.569(3)$ |
| Mg 1 | -Mg 2 | $2.986(5)$ |  | -Y | $3.633(1)$ |
|  | -2 Mg 2 | $3.188(4)$ |  | -3 Mg 1 | $3.697(3)$ |
|  | -M | $3.203(3)$ |  |  |  |
|  | -2 Mg 1 | $3.302(3)$ | Mg 2 | -Mg 2 | $2.878(4)$ |
|  | -4 Mg 1 | $3.365(3)$ |  | -Mg 1 | $2.986(4)$ |
|  | -2 Mg 2 | $3.386(4)$ |  | -2 Mg 2 | $3.027(4)$ |
|  | -M | $3.697(3)$ |  | -2 Mg 1 | $3.188(3)$ |
|  |  |  |  | -2 Mg 1 | $3.386(4)$ |
|  |  |  |  | -2 M | $3.463(3)$ |
|  |  |  |  | -Y | $3.470(3)$ |
|  |  |  |  | -M | $3.569(3)$ |



Fig. 1. Coordination polyhedron of $Y$ (top) and of the mixed site $M$ (bottom).

Table 2
Refined structural parameters for $\mathbf{M g}_{6.8} Y$

| Atom | Site | $x$ | $y$ | $z$ | $U_{\text {cq }}\left(10^{-2} \AA^{2}\right)$ | $U_{11}$ | $U_{22}$ | $U_{33}$ | $U_{12}$ | $U_{13}$ |
| :--- | :--- | :--- | :--- | :--- | :--- | :--- | :--- | :--- | :--- | :--- |
| Y | $2 a$ | 0 | 0 | 0 | $0.97(5)$ | $0.97(9)$ | $U_{11}$ | $U_{11}$ | 0 | 0 |
| M | $8 c$ | $0.3136(1)$ | $x$ | $x$ | $1.37(4)$ | $1.37(7)$ | $U_{11}$ | $U_{11}$ | $0.20(6)$ | $U_{12}$ |
| Mg 1 | $24 g$ | $0.3576(2)$ | $x$ | $0.0357(3)$ | $2.1(1)$ | $2.4(2)$ | $U_{11}$ | $1.5(2)$ | $-0.1(2)$ | $0.6(1)$ |
| Mg 2 | $24 g$ | $0.0905(2)$ | $x$ | $0.2807(3)$ | $1.8(1)$ | $1.9(1)$ | $U_{11}$ | $1.7(2)$ | $-0.7(2)$ | $-0.33(9)$ |

$\bar{M}=0.679(9) \mathrm{Y}+0.321(-) \mathrm{Mg} \Rightarrow$ refined composition: $\mathrm{Mg}_{6.80(7)} \mathrm{Y}$. Space group $I \overline{4} 3 \mathrm{~m}$ (no. 217); Pearson code cl58. $a=11.2507(2) \AA$, $V=1424.09(8) \AA^{3}, D_{x}=2.2 \mathrm{~g} \mathrm{~cm}^{-3}$.
atom distribution in magnesium-poor crystals is desirable.

Finally, it is worth pointing out that the two sites occupied by yttrium have very similar atom coordinations which can be described as 16 -vertex FrankKasper polyhedra (see Fig. 1). The average bond distances involving these sites are $3.51 \AA(\mathrm{Y}$ on $2 a)$ and $3.49 \AA$ ( M on $8 c$ ). No M-M contacts (M: mixed site) are present in the structure.

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